

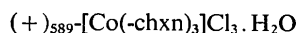
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**The crystal structure of (+)<sub>589</sub>-tris-[-(-)-*trans*-1,2-diaminocyclohexane]cobalt(III) chloride monohydrate ('*ob*' isomer). Corrigenda.** By A. KOBAYASHI, F. MARUMO and Y. SAITO, *The Institute for Solid State Physics, The University of Tokyo, Roppongi-7, Minato-ku, Tokyo 106, Japan*

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The atomic parameters of O(1) in the paper by Kobayashi, Marumo & Saito [*Acta Cryst.* (1972). B28, 2709–2715] are incorrect and the last line of Table 2(a) should read O(1) 2968 (6) 4991 (13) 7042 (10) 94 (6) 194 (11) 254 (16) 40 (12) 90 (8) 9 (20).

An incorrect population factor was used for O(1) in our recent paper on the crystal structure of



(Kobayashi, Marumo & Saito, 1972). The oxygen atoms are randomly distributed on the fourfold general position with population of 0.5. The last line in Table 2(a) should read as in the above abstract.

There is no significant change in the positional and thermal parameters of the other atoms. The final *R* value is 0.031.

#### Reference

KOBAYASHI, A., MARUMO, F. & SAITO, Y. (1972). *Acta Cryst.* B28, 2709–2715.

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**The structures of fluorides. V. The *x*-parameter in NiF<sub>2</sub>.** By J. C. TAYLOR and P. W. WILSON, *A. A. E. C. Research Establishment, Lucas Heights, Australia 2232*

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A least-squares refinement with 22 integrated intensities from a neutron powder pattern of NiF<sub>2</sub> gave an unweighted *R* index on *F*<sup>2</sup> of 3.5% and a fluorine *x* parameter 0.3052 (6). The octahedron was found to be undistorted, confirming that rutile-type fluorides as well as oxides cannot be completely described by a simple Born model.

The observed distortions of octahedra found in rutile-type AB<sub>2</sub> oxides and fluorides are of interest because of their correlation with theoretical predictions. These distortions can be expressed in terms of the difference between the lengths of the 4 coplanar bonds (*l*<sub>1</sub>) located in the (110) plane and the 2 axial bonds (*l*<sub>2</sub>) normal to (110). A Born model predicts *l*<sub>1</sub> > *l*<sub>2</sub>. Baur & Khan (1971) found with X-ray diffraction that, while CrO<sub>2</sub>, RuO<sub>2</sub>, SnO<sub>2</sub>, OsO<sub>2</sub>, PbO<sub>2</sub>, MgF<sub>2</sub>, MnF<sub>2</sub>, FeF<sub>2</sub>, CoF<sub>2</sub>, NiF<sub>2</sub> and ZnF<sub>2</sub> have *l*<sub>1</sub> > *l*<sub>2</sub>, SiO<sub>2</sub>, TiO<sub>2</sub> and GeO<sub>2</sub> have *l*<sub>1</sub> < *l*<sub>2</sub>. In view of this, they pointed out that the Born model was probably inadequate. Jahn–Teller distortions are only likely in FeF<sub>2</sub> (Knox, 1961).

While studying fluorides present in fluorine generators, we measured the *x*<sub>F</sub> parameter of NiF<sub>2</sub> at 21 °C by powder neutron diffraction. This technique does not have the problems of absorption and extinction encountered in the X-ray method. Pure NiF<sub>2</sub> was prepared by heating NiCl<sub>2</sub> in F<sub>2</sub> at 275 °C in a flow reactor (Brauer, 1963), and removing absorbed F<sub>2</sub> (Watanabe & Takashima, 1971) by pumping. The pattern was collected on HIFAR, the AAEC research reactor, by the elastic diffraction technique (Caglioti, 1970) with λ = 1.075 Å. *x*<sub>F</sub>, *B*<sub>Ni</sub>, and *B*<sub>F</sub> where *B* is the isotropic Debye–Waller factor, were determined by a least-squares refinement of 22 integrated intensities, the function minimized being  $\sum_i [w_i(\sum_j JF_o^2 - \sum_j JF_c^2)^2]$ , where *i* is a peak with *j* components each with multiplicity *J*.

The final discrepancy index,  $R = \frac{\sum_i (\sum_j JF_o^2 - \sum_j JF_c^2)}{\sum_i JF_o^2}$

Table 1. Observed and calculated integrated intensities in the neutron powder pattern of NiF<sub>2</sub> at 21 °C

<i>hkl</i>	$\sum JF_o^2$	$\sum JF_c^2$	$\sigma$
110	21.1	21.0	0.4
101	12.5	13.4	0.4
020	0.3	0.6	0.3
111	29.3	28.7	0.6
120	11.6	13.1	0.5
121	103.1	102.6	1.5
220	39.6	41.3	1.1
002	33.1	32.6	1.2
130	15.1	14.6	1.0
221	5.9	5.7	0.9
112 } 031 }	151.6	148.0	2.3
131 }	15.5	15.0	1.3
230 } 022 }	4.9	4.5	1.1
122 }	24.9	23.3	1.4
231 }	5.1	5.4	1.1
040 }	21.4	20.4	1.6
140 } 222 }	102.3	101.0	2.5
330 }	42.1	44.4	1.9
132 } 141 }	72.7	75.6	2.5
240 } 331 }	24.5	22.1	1.9
103 }	12.7	11.4	1.7
232 } 241 }	48.1	51.6	2.3
113 }			

$\sum_i \sum_j JF_{ij}^2$  was 0.035. The observed and calculated intensities are given in Table 1. The thermal parameters found were  $B_{Ni} = 0.45$  (8)  $\text{\AA}^2$  and  $B_F = 0.56$  (9)  $\text{\AA}^2$ . These may be low if thermal diffuse scattering (TDS) is present, but TDS normally has little effect on positional parameters (Cooper, 1970). The correlation coefficients involving  $x_F$  and the scale factor,  $B_{Ni}$  and  $B_F$  were low ( $-0.007$ ,  $0.008$  and  $0.084$  respectively).

Whereas the previous X-ray results indicated a slight distortion, the neutron refinement gives an undistorted octahedron in  $\text{NiF}_2$ . The present values of  $l_1$  and  $l_2$  are similar to the Ni-F distance of  $2.006 \text{\AA}$  in the perovskite  $\text{KNiF}_3$ , where the octahedron is regular by symmetry (Knox, 1961). The deviation from the Born model for  $\text{NiF}_2$  is greater than in the previous work, lending further support for the conclusion of Baur & Khan (1971) that the simple Born model alone may not explain the observations.

The results of the present neutron and previous X-ray work are given below:

Study	$x_F$	$l_1(\text{\AA})$	$l_2(\text{\AA})$
Present work (neutron)	0.3052 (6)	2.005 (3)	2.007 (4)
Baur & Kahn (1971) (X-ray)	0.3012 (13)	2.022 (6)	1.981 (9)

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**Cell constants of two hexachloropalladates.** By J. M. ADAMS, *Edward Davies Chemical Laboratories, Aberystwyth, Cardiganshire, Wales*

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The unit-cell constants of potassium and ammonium hexachloropalladate have been redetermined as  $9.7066$  (3) and  $9.8222$  (4)  $\text{\AA}$  respectively at  $20^\circ\text{C}$ .

Both potassium and ammonium hexachloropalladates have the  $\text{K}_2\text{PtCl}_6$  structure (Wyckoff, 1965), with space group  $Fm\bar{3}m$  (No. 225). Sharpe (1953) determined their lattice constants to be  $9.74$  (1) and  $9.84$  (1)  $\text{\AA}$ . Bell, Hall & Wates (1966) redetermined the lattice constant of the ammonium salt to an accuracy of  $0.01 \text{\AA}$ . Greater accuracy has now been obtained by the use of relatively standard equipment and techniques.

Samples were obtained from FLUKA AG. Analysis results were:

$\text{K}_2\text{PdCl}_6$	26.8% Pd calc.,	26.7% Pd found;
$(\text{NH}_4)_2\text{PdCl}_6$	30.0% Pd calc.,	30.6% Pd found.

The positions of the peaks for angles ( $2\theta$ ) from  $2^\circ$  to  $150^\circ$  were obtained with unfiltered copper radiation on a Philips PW1050 diffractometer. Wavelengths of the different X-ray lines used were:

Cu $K\bar{\alpha}$	1.54178 $\text{\AA}$
Cu $K\alpha_1$	1.54051
Cu $K\alpha_2$	1.54433
Cu $K\beta$	1.39217 .

For all the reflexions arising from Cu  $K\alpha$ , a graph of  $d_{\text{obs}}$  against  $\cot(\theta_{\text{obs}})$  was drawn adapting the method of Bracher & Small (1967) to eliminate zero errors in  $\theta$ . The  $\theta$  values for all of the reflexions were corrected and used in the program *CELFIT* of Bracher (1967) to refine the cell parameter. The parameters obtained for the two compounds at  $20^\circ\text{C}$  are:

$\text{K}_2\text{PdCl}_6$	9.7066 (3) $\text{\AA}$
$(\text{NH}_4)_2\text{PdCl}_6$	9.8222 (4) $\text{\AA}$ .

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